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# Introduction

Electric cars are back on the mass market as an environmentally friendly mode of personal transportation. However, current consumers are cautious and have voiced concerns about long charging times which limits autonomy.1 Importantly, speed of charge is related to slow transport kinetics inside the battery. This is, in turn, often associated with the kinetics of the lithium insertion/deinsertion reaction in the ceramic electroactive solid, since solid-state diffusion generally is slow compared to the liquid and the gas phase.<sup>2</sup> Specifically, LiFePO<sub>4</sub> though superior in many aspects to other positive electrode materials, is criticized for low electronic conductivity and bulk lithium diffusivity.3 Yet, in model systems, LiFePO<sub>4</sub> electrode materials repeatedly exhibit a much larger diffusion coefficient (around  $10^{-8}$  cm<sup>2</sup> s<sup>-1</sup>)<sup>4,5</sup> than in LiFePO<sub>4</sub> powders designed for application (around 10<sup>-14</sup> cm<sup>2</sup> s<sup>-1</sup>).<sup>2</sup> Moreover, a limited number of experimental results have recently shown that commercially relevant LiFePO<sub>4</sub> might be capable of much higher charge/ discharge rates, than previously thought. For example, Ceder and co-workers published a controversial<sup>6</sup> study on a modified LiFePO<sub>4</sub> material with non-stoichiometric composition and amorphous surface layer, with which approximately 75% of the theoretical capacity could be achieved within a one minute discharge.7 Similarly an electrochemical single particle study by Munakata and co-workers showed about 75% of the initial capacity at a one minute discharge.8

# Ultrafast charging of LiFePO<sub>4</sub> with gaseous oxidants under ambient conditions<sup>†</sup>

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Lithium iron phosphate is a lithium-ion battery positive electrode material with widespread use, as well as, unusually complex redox chemistry. Here we report on the discovery of a direct gas–solid delithiation reaction. Unique to this reaction, in addition to the lack of solvent, is remarkably fast kinetics. *In situ* X-ray diffraction, corroborated by elemental analysis, shows for the first time that LiFePO<sub>4</sub> bulk diffusion supports nearly complete delithiation/charging of carbon coated LiFePO<sub>4</sub> micropowder at ambient temperature in less than 60 seconds.

At the same time, industry is concerned with the stability of  $LiFePO_4$  in ambient atmosphere, as this is of great importance to the storage and handling of commercial  $LiFePO_4$  during production. For example, complete transformation of  $LiFePO_4$  into the NASICON analogue  $Li_3Fe_2(PO_4)_3$  and hematite was observed during exposure to air at 300 °C and above,<sup>9</sup> whereas at temperatures below 120 °C, in humid air, the formation of hydroxide containing compounds has been reported.<sup>10,11</sup> Yet, it remains unclear, why lithium is *not* extracted under oxidative aging conditions, even though this is the dominating mode of oxidation in solution.

To address this question of different reaction modes in air compared to electrolyte, and to shed light on the lithium transport kinetics of these reactions, we have examined the impact of different gaseous oxidants on commercial LiFePO<sub>4</sub>. Surprisingly, we have found that even though exposure to  $O_2/O_3$ did *not* significantly alter the LiFePO<sub>4</sub> materials, NO<sub>2</sub> consistently delithiates LiFePO<sub>4</sub> completely within a short period of time according to the following reaction:

$$LiFePO_4(s) + 2NO_2(g) \rightarrow FePO_4(s) + LiNO_3(s) + NO(g)$$

This reaction differs significantly from previous oxidative delithiation transformations, as it does not include a liquid phase that can solvate the lithium ion and transport it away from the surface as the reaction progresses. More importantly, it exhibits unseen fast reaction rates for commercial LiFePO<sub>4</sub> materials.

### **Results and discussion**

#### Characterization

To confirm, that this reaction indeed is comparable to electrochemical charging, the solid reaction products have been characterized using attenuated total reflectance infrared

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spectroscopy (ATR-FTIR), X-ray diffraction (XRD), electrochemical cycling and transmission electron microscopy.

After treatment of carbon-coated LiFePO<sub>4</sub> (C-LiFePO<sub>4</sub>) micropowders with NO<sub>2</sub>, evidence of delithiation was given by LiNO<sub>3</sub> and *heterosite* FePO<sub>4</sub> as identified by ATR-FTIR and XRD (Fig. 1c and f). Specifically, the LiNO<sub>3</sub> gives rise to FTIR bands at 737, 838, 1072, 1135 cm<sup>-1</sup> and a broad feature between 1300 and 1500 cm<sup>-1</sup>, as well as a number of shoulders, which appear in addition to the standard *heterosite* FePO<sub>4</sub> spectrum (Fig. 1f).<sup>15</sup>

The completeness of the delithiation was confirmed by atomic emission spectroscopy (AES): 100  $\pm$  3% lithium was extracted, while 2  $\pm$  1% lithium remained in the washed FePO<sub>4</sub> sample.

As the use of aggressive oxidants to delithiate LiFePO<sub>4</sub> might lead to the formation of non-crystalline by-products or particle dissolution, high-resolution transmission electron micrographs of nano-sized carbon-free LiFePO<sub>4</sub> were recorded to assess structural integrity of the reaction product. From Fig. 2, it is clear that the overall shape, size and appearance of the particles remain unaltered. Furthermore, FePO<sub>4</sub> particles remain crystalline while a salt layer forms non-uniformly on the surface, accumulating in gaps and contact points. X-Ray photoelectron spectroscopy reveals a nitrogen containing compound on the material surface. Fig. 3 shows the N 1s peak at 406.7 eV, lying in



**Fig. 1** Crystallographic and chemical analysis of the reaction product of C-LiFePO<sub>4</sub> with O<sub>3</sub> and NO<sub>2</sub>. X-Ray diffractograms (a–c) and ATR FTIR spectra (d–f) of pristine C-LiFePO<sub>4</sub> (a and d), O<sub>3</sub> exposed C-LiFePO<sub>4</sub> (b and e) and NO<sub>2</sub> oxidized C-LiFePO<sub>4</sub> (c and f). The symbols mark the location of strong reflections according to literature crystallographic data.<sup>12-14</sup>



**Fig. 2** TEM images of LiFePO<sub>4</sub> before oxidation (a and c) and after oxidation with nitrogen dioxide (b and d). HRTEM images show crystallinity of particles up to the surface before oxidation (c) and an amorphous surface layer after oxidation (d).

between the values reported for  $AgNO_3^{16}$  and  $NH_4NO_3^{17}$  thus suggesting the presence of LiNO<sub>3</sub> on the surface. The same references report O 1s peaks at 532.3 and 532.5 eV, which compares well to the observed component at 532.4 eV.

The electrochemical activity of oxidized C-LiFePO<sub>4</sub> was assessed in research coin cell batteries, assembled with great care to avoid any accidental short-circuit. Electrochemical testing was initiated in discharge mode, without prior charging. This first discharge (Fig. 4a) indicates a stable potential plateau



Fig. 3  $\,$  XPS analysis of the nitrogen and oxygen 1s peaks of the oxidized sample confirms the presence of LiNO<sub>3</sub> at the surface.



**Fig. 4** (a) First-cycle discharge curve (rate C/10) and (b) cycling performance (rate C/2) of oxidized, washed and dried C-LiFePO<sub>4</sub> confirm complete oxidation and retention of electrochemical activity of the oxidized material.

around 3.4 V vs. Li/Li<sup>+</sup> and practical capacity of 165 mA h g<sup>-1</sup> (theoretical capacity: 170 mA h g<sup>-1</sup>). Combined with the cycling stability over 50 cycles (Fig. 4b), this indicates that the material retains its electrochemical properties and is not damaged by the aggressive delithiation.

As mentioned above, LiFePO<sub>4</sub> was also exposed to ozone. This gas did not lead to a significant alteration of the starting material, *i.e.* the bulk *olivine* structure remains intact, as observed in XRD, and further confirmed by only very minor changes to the ATR-FTIR spectrum (Fig. 1b and e).

#### Thermodynamics of LiFePO<sub>4</sub> delithiations with gases

It is clear, that the key to the observed differentiated reaction behaviours lies within the nature of the oxidant. Table 1 summarizes some reaction Gibbs free energies for delithiation reactions of LiFePO<sub>4</sub> with different oxidizing gases using actual reaction conditions for NO<sub>2</sub>,  $Cl_2$  and O<sub>3</sub> oxidations, and ambient conditions for O<sub>2</sub>. The thermodynamic discussion of oxidation pathways of LiFePO<sub>4</sub> with gases may further be extended to the

 Table 1
 Gibbs free energies of delithiation reactions under reaction/ambient conditions<sup>a,18,19</sup>

Reaction	$\Delta_{ m R}G/ m kJ~mol^{-1}$
$LiFePO_4 + \frac{1}{2}Cl_2 \rightarrow LiCl + FePO_4$	-53
$LiFePO_4 + 2NO_2 \rightarrow LiNO_3 + FePO_4 + NO_4$	-65
$LiFePO_4 + \frac{1}{2}O_3 \rightarrow \frac{1}{2}Li_2O + FePO_4 + \frac{1}{2}O_2$	-19
$\text{LiFePO}_4 + \frac{1}{2}\text{O}_3 + \frac{1}{2}\text{H}_2\text{O} \rightarrow \text{LiOH} + \text{FePO}_4 + \frac{1}{2}\text{O}_2$	-60
$LiFePO_4 + \frac{1}{4}O_2 \rightarrow \frac{1}{2}Li_2O + FePO_4$	+51
$LiFePO_4 + \frac{1}{4}O_2 + \frac{1}{2}H_2O \rightarrow LiOH + FePO_4$	+11
$\mathrm{LiFePO_4} + \frac{1}{4}\mathrm{O_2} + \frac{1}{2}\mathrm{CO_2} \rightarrow \frac{1}{2}\mathrm{Li_2CO_3} + \mathrm{FePO_4}$	-38

 $^a$  20 °C, 20.9% O2, 0.035% of CO2, and 70% rel. humidity were assumed ambient conditions.

extraction of iron ions from, or introduction of oxygen into the LiFePO<sub>4</sub> structure. However, molecular modelling shows, that those ions are strongly bound to their lattice site in the LiFePO<sub>4</sub> structure, compared to a more mobile lithium.<sup>18</sup> Bulk diffusion kinetics should hence favour delithiation reactions. Delithiation of LiFePO<sub>4</sub> with O<sub>3</sub> and O<sub>2</sub> in the presence of CO<sub>2</sub> is thermodynamically possible with free energies down to -60 kJ mol<sup>-1</sup> at standard conditions depending on the pathway.<sup>19</sup> This delithiation is *not* observed, suggesting surface kinetics are responsible for its inhibition *e.g.* surface localized species may block the reaction. Given that ozone is a strongly oxidizing allotrope of oxygen, the reaction products of O<sub>2</sub>/LiFePO<sub>4</sub> and O<sub>3</sub>/LiFePO<sub>4</sub> may be quite similar, thus potentially yielding new information on the dry air aging mechanism of LiFePO<sub>4</sub>.

#### Kinetics

In an attempt to quantify the exceptionally high reaction rate, *in situ* time-resolved XRD was performed. Fig. 5 shows evidence of



**Fig. 5** (a) Time-resolved XRD during delithiation of C-LiFePO<sub>4</sub> by NO<sub>2</sub> gas as a greyscale map. The initial and final diffractograms are displayed at the top and bottom, respectively; t = 0 marks the time of gas injection. (b) Composition of the mixture LiFePO<sub>4</sub>/FePO<sub>4</sub>. The composition was determined from time-resolved XRD by integration and normalization to the corresponding theoretical intensity of the LiFePO<sub>4</sub> reflection at 30° 2 $\theta$  and the FePO<sub>4</sub> reflection at 31° 2 $\theta$  (based on a Cu-K $\alpha$  anode X-ray source).

complete delithiation of C-LiFePO<sub>4</sub> particles of 590 nm average diameter within significantly less time than one minute. The rate of delithiation was also confirmed by AES with  $61 \pm 14\%$ delithiation at 30 s and  $94 \pm 4\%$  delithiation at 60 s. This translates into a LiFePO<sub>4</sub> charge to 160 mA h g<sup>-1</sup> within one minute. Importantly, the reaction temperature peaked at 29.7 °C  $\pm$  1.1 °C, thus excluding any major thermal increase of the kinetics resulting from the exothermic nature of the reaction. Moreover, preliminary tests using C-LiFePO<sub>4</sub> and Cl<sub>2</sub> have shown similar kinetics. LiFePO<sub>4</sub> nanopowder samples of approximately 200 nm average particle diameter with and without carbon coating have been studied as well. Regardless of the presence or absence of coating, these showed reaction rates that were too fast to be captured within the 15 seconds time resolution of this conventional X-ray diffraction set-up.

For comparative purposes, our data provides a *lower limit* on the apparent diffusion coefficient of about  $3.1 \times 10^{-11}$  cm<sup>2</sup> s<sup>-1</sup>, using a one-dimensional pure diffusion model, as has been done in previous electrochemical studies.<sup>20</sup> As such, this study shows that the rates provided by Ceder *et al.*<sup>7</sup> and Munakata *et al.*<sup>8</sup> are entirely feasible *provided* that the removal of electrons from the particle surface is sufficiently fast.

# Conclusion

Unique to the gas reaction discovered here, is the delithiation of LiFePO<sub>4</sub> at high speed without the presence of a liquid. In situ X-ray diffraction corroborated by elemental analysis provides proof that LiFePO<sub>4</sub> bulk kinetics supports a charge to 160 mA h  $g^{-1}$  in less than 60 seconds under ambient conditions. This finding has been confirmed with two LiFePO4 materials resulting from different synthesis routes regardless of the presence or absence of carbon coating. The reaction is comparable to the electrochemical process in so far as the resulting FePO<sub>4</sub> is indistinguishable from electrochemically delithiated Li<sub>0</sub>FePO<sub>4</sub> and the thermodynamic driving force corresponds to a charge to 4.1 V vs. Li/Li<sup>+</sup>. It provides thus new possibilities to study the delithiation mechanism of LiFePO<sub>4</sub> in situ and ex situ. As such, XRD and TEM studies are currently underway. The findings further disprove the paradigm of slow lithium bulk diffusion in LiFePO<sub>4</sub>.

In conclusion, the presented data suggest that developing  $LiFePO_4$  materials with improved bulk lithium diffusivity will not improve rate capabilities of the derived lithium-ion batteries. Instead, electrode design, electronic conductivity and surface kinetics should be the focus of continued research.

# Experimental

Micro-sized carbon coated LiFePO<sub>4</sub> (C-LiFePO<sub>4</sub>, *US Pat.*, 7,457,018) and carbon-free nano-LiFePO<sub>4</sub> (*US Pat.*, 7,807,121 B2) were donated by Clariant (Canada) Inc. (former Phostech Lithium Inc).

C-LiFePO<sub>4</sub> (chemical and crystallographic analysis) and carbon-free nano-LiFePO<sub>4</sub> (used for XPS) samples were exposed to nitrogen dioxide and ozone gas, respectively, for at least 30 minutes. For chemical quantification of the oxidation,

oxidized samples were washed in water and filtered. FePO<sub>4</sub> was subsequently dissolved in conc. HNO<sub>3</sub>. Wash water and dissolved FePO<sub>4</sub> were analysed by AES. To achieve time resolution, the oxidation was stopped after different exposure times by replacing NO<sub>2</sub> gas with a stream of dry air. TEM samples were prepared by depositing carbon-free nano-LiFePO<sub>4</sub> onto lacey carbon nickel grids from a suspension in acetonitrile. Selected sample covered grids were exposed to NO<sub>2</sub> gas before analysis in the TEM. Carbon-coated nano-LiFePO<sub>4</sub> shows the same characteristics.

The electrodes for battery testing were produced by coating 85 wt% washed, completely oxidized LiFePO<sub>4</sub>, 6 wt% PVDF binder and 9 wt% carbon additive on to a carbon-coated aluminium foil. The battery contained a metallic lithium negative electrode and LiPF<sub>6</sub> in 1:1 ethylene carbonate and dimethyl carbonate mixture electrolyte.

Time-resolved X-ray diffraction was performed using a flow of NO<sub>2</sub> gas below a filter paper on which LiFePO<sub>4</sub> was fixed. X-ray access to the XRD cell was enabled through a Kapton window. In similar experiments, the peak temperature of LiFePO<sub>4</sub> during NO<sub>2</sub> oxidation was recorded, using an infrared thermometer and confirmed in independent experiments with a thermocouple.

For more experimental details, suppliers and instruments, please see the ESI. $\dagger$ 

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